

Article

Analysis and Prediction of Corrosion of Refractory Materials by Sodium Salts during Waste Liquid Incineration—Thermodynamic Study

Guishi Cheng ¹, Ying Zhao ^{1,*}, Fei Long ², Jiahe Zhang ¹, Tengfei Zhao ¹, Lu Liu ¹,
Xiaoqiang Wang ¹ and Changqing Dong ^{1,3}

¹ National Engineering Laboratory for Biomass Power Generation Equipment, School of New Energy, North China Electric Power University, Beijing 102206, China; 51101890@ncepu.edu.cn (G.C.); 120181100109@ncepu.edu.cn (J.Z.); zhao_012345@163.com (T.Z.); 51102247@ncepu.edu.cn (L.L.); energy@ncepu.edu.cn (X.W.); cqdong@ncepu.edu.cn (C.D.)

² Department of Mechanical and Material Engineering, Queen's University, Kingston, ON K7L 3N6, Canada; long.fei@queensu.ca

³ State Key Laboratory of Alternate Electrical Power System with Renewable Energy Sources, North China Electric Power University, Beijing 102206, China

* Correspondence: zhaoying1989@ncepu.edu.cn; Tel.: +86-010-6177-2060

Received: 24 September 2020; Accepted: 21 October 2020; Published: 23 October 2020



Abstract: Incineration of high-content sodium salt organic waste liquid will corrode the refractory material in the incinerator, causing the refractory material to peel off and be damaged. A thermodynamics method was used to study the thermodynamic properties of three common sodium salts (NaCl, Na₂CO₃ and Na₂SO₄) on the corrosion of refractory materials (MgO·Cr₂O₃, MgO·Al₂O₃, Al₂O₃, MgO and Cr₂O₃). The results determined that MgO has the best corrosion resistance and is not corroded by the three sodium salts. On this basis, the thermodynamic corrosion experiments of NaCl corrosion of magnesium oxide at three temperatures of 600, 1000 and 1200 °C were carried out. Analysis of the corrosion product by X-ray diffraction (XRD) showed no corrosion product formation. Studies have shown that thermodynamic calculation can accurately predict the thermodynamic mechanism of alkali metal corrosion to refractory materials, and MgO is a good anti-alkali metal corrosion refractory material.

Keywords: sodium salt; refractory; corrosion; thermodynamics

1. Introduction

The rapid development of the petrochemical industry has led to an increase in emissions of high-concentration organic waste liquids. This type of wastewater is not only difficult to be decomposed by microorganisms, but also has the potential to induce genetic mutations and toxicity to humans and animals. Incineration is a commonly used method for treating organic waste liquids in industry, especially for some waste liquids with high concentration, complex composition and high heat value. However, the organic waste liquid contains alkali metal salts such as Na₂SO₄, KCl, NaCl, etc., and the acidic waste water is usually neutralized by adding alkaline substances (such as KOH, NaOH) before incineration to reduce the corrosion of the pump and the pipeline by the acidic wastewater, bringing more alkali metal salts to the organic waste liquid. Most alkali metal salts have a low melting point, causing severe corrosion to the refractory layer of the incinerator, affecting the service life of the refractory, and causing great safety hazards and economic losses to the operation of the incinerator [1].

With the advancement of society and the development of science and technology, the performance requirements of refractory materials are getting higher and higher. Many researchers have studied and improved the corrosion resistance of refractories [2–6], thermal shock resistance [7–9], and anti-wear properties [10]. M. Yoshikawa et al. [11] studied the erosion resistance of MgO-based spinel and Al₂O₃-based spinel refractories and found that these two refractories can replace Al₂O₃-Cr₂O₃ refractories with low chromium content and are more suitable for alkali furnaces. P. Prigent et al. [1] studied the sodium corrosion resistance of three types of silica-alumina refractories (andalusite, mullite, and refractory clay) and compared the microstructure of the refractory surface before and after NaF vapour corrosion. J. Moda et al. [12] added MgO and NiO to an alumina refractory to realize Al₂O₃-MgO-NiO refractories and found that a (Mg, Ni)O solid solution was formed in the material and that its ability to resist slag erosion was improved. R. D. Fan et al. [13] found that alumina chrome slag can replace alumina to prepare Al₂O₃-SiC-C trough castables, which remarkably improved oxidation resistance of the sample, but reduced its thermo-mechanical properties and anti-slag corrosion performance. Some researchers have paid attention to the effect of additives on the corrosion resistance of refractory materials. K. Igabo et al. [14] added ZrO₂ to MgO-Al₂O₃ refractories and studied the effect of the addition on the performance of the refractories. P. Gehre et al. [15] reported that adding spinel materials, spinel-rich cements, and calcium aluminate cement to the spinel-containing alumina lining of steel ladles could significantly improve the slag corrosion resistance.

There are many studies concerning the performance of refractories used in the steelmaking process [16–19], but the thermodynamic research on sodium salt corrosion of refractory materials is limited. D. Gregurek et al. [20] used Factsage software to carry out thermodynamic simulation on the corrosion of refractory lining caused by slag and CuO. The research results can provide a theoretical basis for the wear of refractories in copper anode furnaces. J. Stjernberg et al. [21] used XRD, electron microscopy and spectroscopy (Quantitative Evaluation of Minerals by Scanning Electron Microscopy (QEMSCAN) and Scanning Electron Microscopy (SEM)) characterization and thermodynamic kinetics to study the erosion of mullite based refractory bricks by alkali metals. R. Huang et al. [22] calculated the erosion of four refractory materials (i.e., magnesia carbon brick, burned magnesite brick, SiC castable, and corundum castable) by titanium slag using FactSage software. The results found that silicon carbide castable has the best corrosion resistance among the four refractories, because it can interact with Ti slag to form TiC with a high melting point, which can prevent Ti slag from penetrating more deeply into refractories.

Based on the thermodynamic study of refractory corrosion, this paper compares the corrosion resistance of five refractories under different sodium salts, providing a theoretical basis and data support for further research on alkali metal corrosion resistance of refractories.

2. Thermodynamic Calculations

Thermodynamic simulation is based on the version 7.2 of FactSageTM software for thermodynamic equilibrium calculations. In this paper, the components of refractory materials such as MgO-Cr₂O₃, MgO-Al₂O₃, Al₂O₃, MgO and Cr₂O₃ are selected as research objects. The corrosion mechanism of Na₂CO₃, Na₂SO₄ and NaCl on refractory materials at different temperatures (600 °C–1200 °C with a temperature step of 100 °C) was investigated. The refractory materials components of MgO-Cr₂O₃, MgO-Al₂O₃, Al₂O₃, MgO and Cr₂O₃ and 3-sodium salts of 1mol were input in the table of the FactSageTM software for the calculation. In this paper, the Equilib module of FactSageTM 7.2 software was used, which is suitable for calculating the concentration of various species when the reaction of a given element or compound reaches chemical equilibrium.

3. Thermodynamic Experimental Procedure

Similar to our previous work [23], the experiment was conducted in a high-temperature tube furnace (Figure 1). A 1:1 molar ratio of NaCl and MgO was put into a corundum boat and full

mechanical mixing was carried out. The rest of experimental and data analysis procedures were all the same as previously reported [23].

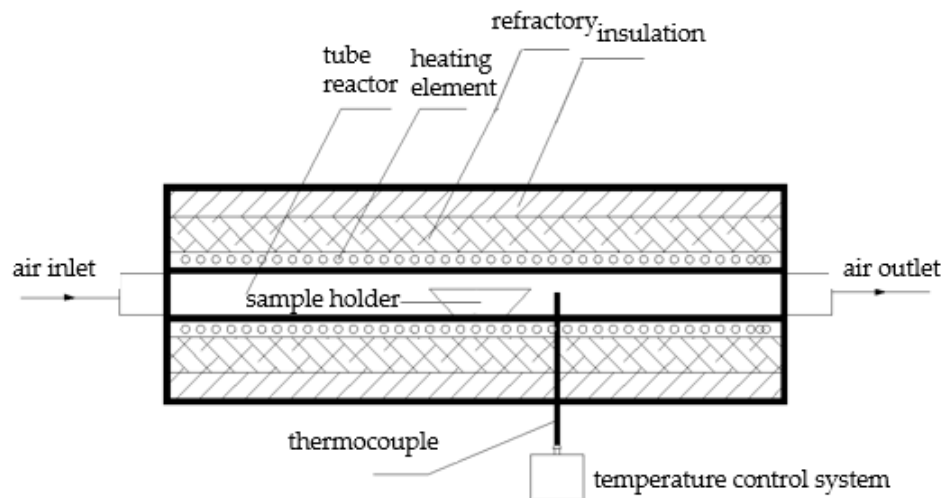


Figure 1. Schematic diagram of experiment setup.

4. Results and Discussion

4.1. Corrosion Effect of Na_2CO_3 on Refractory Materials

Figure 2 illustrates that the Gibbs free energy values for the reactions of Na_2CO_3 with five refractory materials are all negative. As the temperature increases, the Gibbs free energy value decreases, indicating that with the increase in temperature, the corrosion tendency of refractories by Na_2CO_3 becomes more obvious. It can also be obtained from the figure that the order of the tendency of the five refractories to be corroded by Na_2CO_3 is $\text{MgO} \cdot \text{Al}_2\text{O}_3 > \text{MgO} \cdot \text{Cr}_2\text{O}_3 > \text{Al}_2\text{O}_3 > \text{Cr}_2\text{O}_3 > \text{MgO}$.

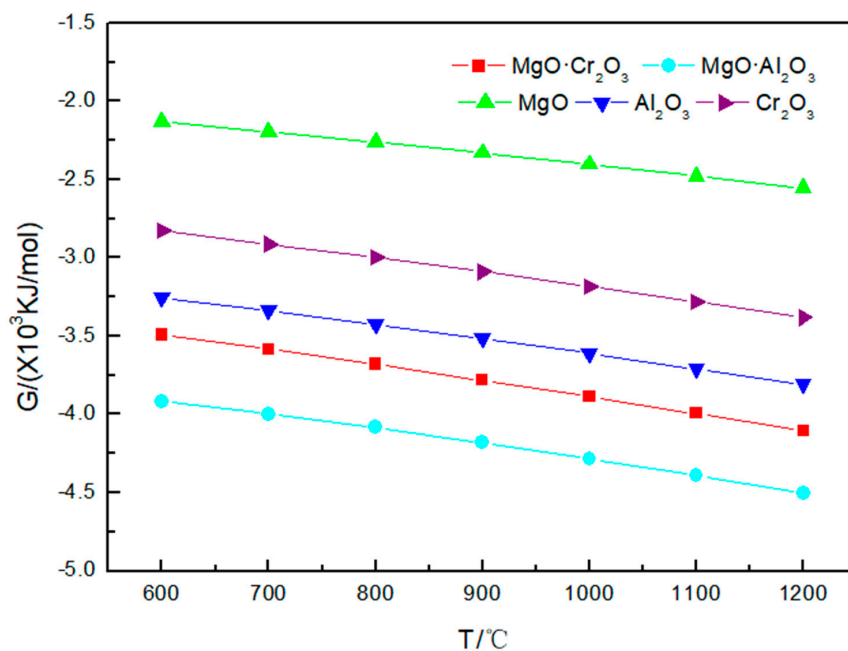


Figure 2. Gibbs free energy of the five corrosion reactions between Na_2CO_3 and refractory materials at 600–1200 °C.

In general, MgO-based refractory materials are basically free from corrosion reaction in the temperature range below 1200 °C, because MgO ions carry two charges and the radii of oxygen ions and magnesium ions are relatively small, which causes magnesium oxide to have large lattice energy, a high melting point and stable properties [24].

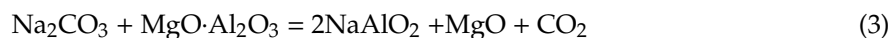
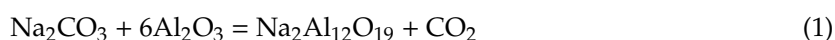
The corrosion products of refractory materials corroded by sodium carbonate and the corrosion reaction equations deduced therefrom are shown in Table 1 shows that, in addition to MgO, other refractories are corroded by sodium carbonate at 600 °C.

Table 1. Predicted corrosion products and reaction equations of Na₂CO₃ on refractory materials at 600–1200 °C.

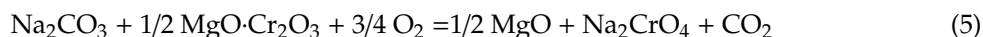
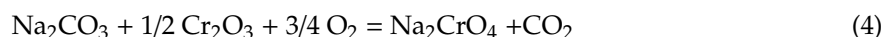
Species	Temperature (°C)			
	600	700	800	900–1200
Al ₂ O ₃	0.16667 mol Na ₂ Al ₁₂ O ₁₉ Equation (1)		2 mol NaAlO ₂ Equation (2)	
MgO		-		
Cr ₂ O ₃		1 mol Na ₂ CrO ₄ Equation (4)		
MgO·Cr ₂ O ₃		1 mol Na ₂ CrO ₄ , 0.5mol MgO Equation (5)		
MgO·Al ₂ O ₃	0.00962 mol NaAlO ₂ , 0.00481 mol MgO Equation (3)	0.08292 mol NaAlO ₂ , 0.04146 mol MgO Equation (3)	0.53843 mol NaAlO ₂ , 0.26922 mol MgO Equation (3)	2 mol NaAlO ₂ , 1 mol MgO Equation (3)

“-”: The amount of corrosion products is less than 10⁻⁴ mol.

The reaction products and reaction degree of Al₂O₃ and MgO·Al₂O₃ corroded by Na₂CO₃ are related to temperature. Al₂O₃ is more easily corroded by Na₂CO₃, and has been completely corroded at 600 °C to generate 0.16667 mol Na₂Al₁₂O₁₉, and 2 mol of NaAlO₂ at 700–1200 °C. At 600–1100 °C, only a small amount of MgO·Al₂O₃ is corroded to form NaAlO₂ and MgO; when the temperature reaches 1200 °C, MgO·Al₂O₃ is completely corroded. The reaction of Al₂O₃ and MgO·Al₂O₃ by Na₂CO₃ corrosion is as follows:



In the reaction of Cr₂O₃ and MgO·Cr₂O₃ by Na₂CO₃, Cr³⁺ is oxidized to Cr⁶⁺ due to the presence of O₂, and the corrosion of Cr₂O₃ and MgO·Cr₂O₃ at 600–1200 °C by Na₂CO₃ is:



In addition, it can be seen from the table that no corrosion product of Na₂CO₃ corroding MgO is found. However, Na₂CO₃ will decompose at 600–1200 °C, so the Gibbs free energy of the MgO·Na₂CO₃ reaction is negative. To sum up, MgO is hard to be eroded by Na₂CO₃.

4.2. Corrosion Reaction of Na₂SO₄ and Refractory Materials

As shown in Figure 3 and Table 2, under 600–1200 °C, neither MgO nor MgO·Al₂O₃ will be corroded by sodium sulfate. A small amount of Al₂O₃ (0.00135 mol) and Cr₂O₃ (0.0001 mol) will be corroded by sodium sulfate at 1000 °C. As the temperature rises to 1200 °C, more Al₂O₃ (0.1425 mol)

and Cr_2O_3 (0.00365 mol) will be corroded by sodium sulfate. The reaction of Al_2O_3 and Cr_2O_3 corroded by sodium sulfate is:

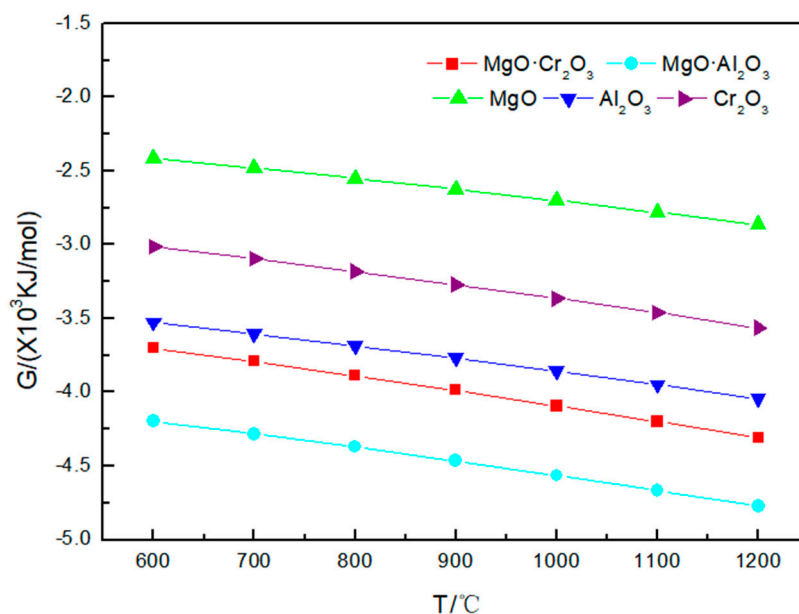
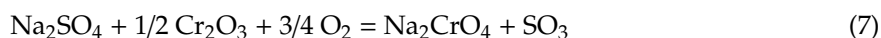
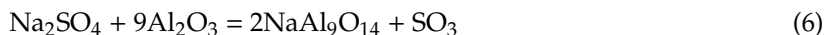


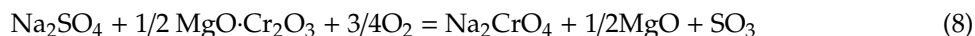
Figure 3. Gibbs free energy of the five corrosion reactions between Na_2SO_4 and refractory materials at 600–1200 °C.

Table 2. Predicted corrosion products and reaction equations of Na_2SO_4 on refractory materials at 600–1200 °C.

Species	Temperature (°C)			
	600–900	1000	1100	1200
Al_2O_3	-	0.0003 mol $\text{NaAl}_9\text{O}_{14}$ Equation (6)	0.0033mol $\text{NaAl}_9\text{O}_{14}$ Equation (6)	0.031672 mol $\text{NaAl}_9\text{O}_{14}$ Equation (6)
MgO			-	
Cr_2O_3	-	0.0002 mol Na_2CrO_4 Equation (7)	0.0013 mol Na_2CrO_4 Equation (7)	0.0073 mol Na_2CrO_4 Equation (7)
$\text{MgO}\cdot\text{Cr}_2\text{O}_3$		-	0.0001 mol Na_2CrO_4 Equation (8)	0.0008 mol Na_2CrO_4 , 0.0004mol MgO Equation (8)
$\text{MgO}\cdot\text{Al}_2\text{O}_3$			-	

"-": The amount of corrosion products is less than 10^{-4} mol.

Similarly, a very small amount of $\text{MgO}\cdot\text{Cr}_2\text{O}_3$ (0.00005 mol) will be corroded by sodium sulfate to form Na_2CrO_4 at 1100 °C. With the temperature rises, even if the temperature rises to 1200 °C, as long as 0.0004 mol of $\text{MgO}\cdot\text{Cr}_2\text{O}_3$ is corroded by sodium sulfate to form Na_2CrO_4 , the reaction of $\text{MgO}\cdot\text{Cr}_2\text{O}_3$ being corroded by sodium sulfate is:



From the above results, it can be seen that the order of corrosion degree of five refractories by Na_2SO_4 from strong to weak is as follows: Cr_2O_3 and $\text{Al}_2\text{O}_3 > \text{MgO}\cdot\text{Cr}_2\text{O}_3 > \text{MgO}\cdot\text{Al}_2\text{O}_3$ and MgO.

4.3. Corrosion Reaction of NaCl and Refractory Materials

The results in Figure 4 show that the Gibbs free energy of the refractory material corroded by sodium chloride has a tendency to change with that of the refractory material with sodium carbonate and sodium sulfate at 600–1200 °C. It can be seen from Table 3 that NaCl is less corrosive to refractory materials. Even at 1200 °C, MgO and MgO·Al₂O₃ are not corroded by sodium chloride; only a small amount of Al₂O₃ (0.00855 mol), Cr₂O₃ (0.0003 mol) and MgO·Cr₂O₃ (0.000005 mol) is corroded by sodium chloride, and the reaction equation is:

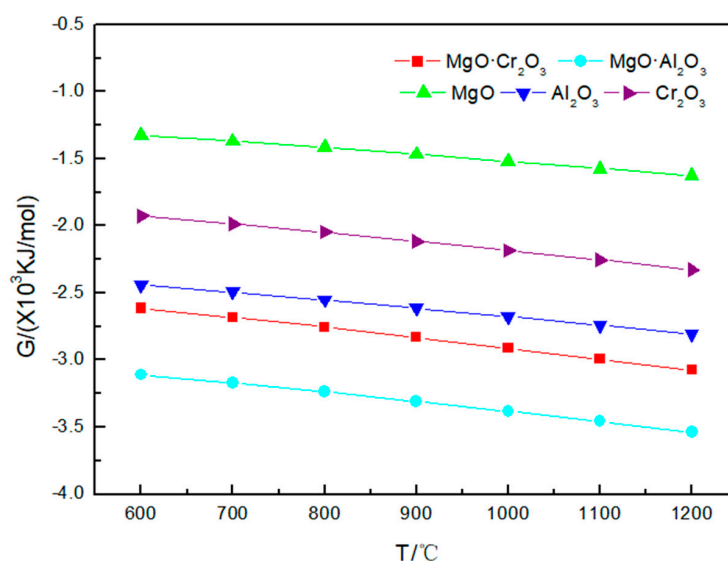
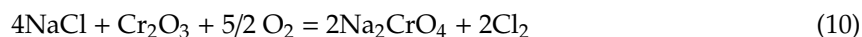
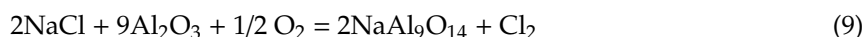


Figure 4. Gibbs free energy of the five corrosion reactions between NaCl and refractory materials at 600–1200 °C.

Table 3. Predicted corrosion products and reaction equations of NaCl on refractory materials at 600–1200 °C.

Species	Temperature (°C)				
	600–800	900	1000	1100	1200
Al ₂ O ₃	-	0.0001 mol NaAl ₉ O ₁₄ Equation (9)	0.0003 mol NaAl ₉ O ₁₄ Equation (9)	0.0008 mol NaAl ₉ O ₁₄ Equation (9)	0.0019 mol NaAl ₉ O ₁₄ Equation (9)
MgO	-				
Cr ₂ O ₃	-	0.0001 mol Na ₂ CrO ₄ Equation (10)	0.0002 mol Na ₂ CrO ₄ Equation (10)	0.0003 mol Na ₂ CrO ₄ Equation (10)	0.0006 mol Na ₂ CrO ₄ Equation (10)
MgO·Cr ₂ O ₃	-				0.0001 mol Na ₂ CrO ₄ Equation (11)
MgO·Al ₂ O ₃	-				

“-”: The amount of corrosion products is less than 10⁻⁴ mol.

It is worth noting that Figures 2–4 shows that the Gibbs free energy of the reaction of MgO·Cr₂O₃ and MgO·Al₂O₃ corroded by sodium carbonate, sodium sulfate and sodium chloride is smaller than that of Al₂O₃ and Cr₂O₃ corroded by sodium carbonate, sodium sulfate and sodium chloride, but,

as can be seen from Tables 1–3, $\text{MgO} \cdot \text{Cr}_2\text{O}_3$ and $\text{MgO} \cdot \text{Al}_2\text{O}_3$ are less susceptible to corrosion by sodium carbonate, sodium sulfate and sodium chloride than Al_2O_3 and Cr_2O_3 . This is because $\text{MgO} \cdot \text{Cr}_2\text{O}_3$ and $\text{MgO} \cdot \text{Al}_2\text{O}_3$ will undergo decomposition reaction at high temperature to generate MgO and Cr_2O_3 , MgO and Al_2O_3 , respectively. This also shows that the trend of corrosion resistance of refractory materials cannot be judged solely by the Gibbs free energy of the reaction, but must be judged comprehensively by combining reaction products.

Therefore, Al_2O_3 and Cr_2O_3 are most easily corroded by sodium carbonate, sodium sulfate and sodium chloride, followed by $\text{MgO} \cdot \text{Cr}_2\text{O}_3$, and $\text{MgO} \cdot \text{Al}_2\text{O}_3$ and MgO , which are the most difficult to be corroded by sodium salts.

4.4. Thermodynamic Experiment of NaCl Corrosion of MgO

As can be seen from Figure 5, XRD patterns show that only MgO and NaCl are present in the samples at temperatures of 600, 1000 and 1200 °C, indicating that MgO has not been corroded by NaCl , and it has good corrosion resistance, which is consistent with thermodynamic calculation results.

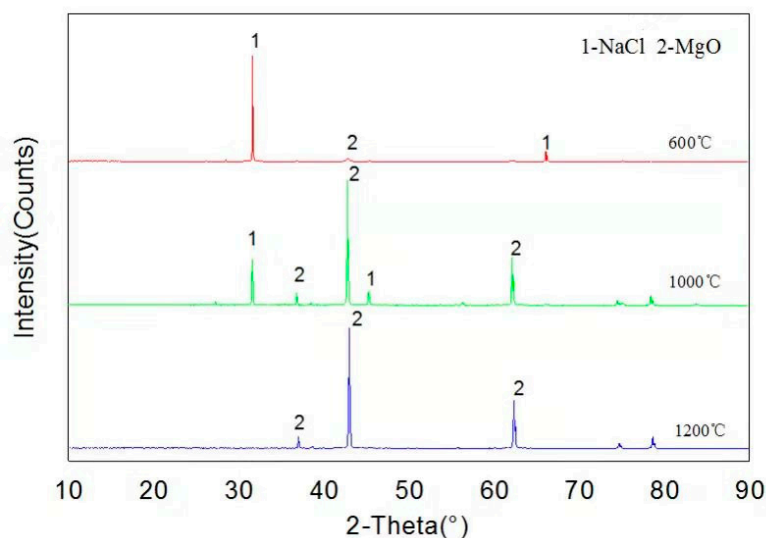


Figure 5. XRD pattern of the reaction products of NaCl and MgO.

Figure 5 shows the XRD spectra of samples after NaCl and MgO high temperature corrosion tests at 600, 1000 and 1200 °C. The results present that only MgO and NaCl were detected in the sample, and no new substance was formed. The experimental results and simulation results by FactSageTM 7.2 software (see Table 3) indicated that MgO is an excellent refractory, which can avoid NaCl corrosion. According to our previous thermodynamic experimental research [23], the thermodynamic model of the software can accurately predict the thermodynamic mechanism of alkali metal corrosion to refractory materials.

5. Conclusions

In this paper, different refractory components ($\text{MgO} \cdot \text{Al}_2\text{O}_3$, $\text{MgO} \cdot \text{Cr}_2\text{O}_3$, Al_2O_3 , Cr_2O_3 and MgO) are corroded by different sodium salts (NaCl , Na_2CO_3 and Na_2SO_4) at different temperatures (600–1200 °C). Thermodynamics studies have obtained the effect of sodium salt type and temperature on the corrosion of refractory materials. By comparison and analysis, the refractory materials with good corrosion resistance are obtained.

1. The temperature has a great influence on the corrosion of refractory materials by sodium salt at 600–1200 °C. Besides MgO and $\text{MgO} \cdot \text{Al}_2\text{O}_3$, the higher the temperature, the stronger the corrosion of refractory materials by sodium sulfate and sodium chloride.

2. Among the five refractory materials, MgO has the best resistance to sodium salt corrosion, followed by MgO·Cr₂O₃ and MgO·Al₂O₃; Cr₂O₃ and Al₂O₃ have the worst resistance to sodium salt corrosion. Due to the presence of O₂, Cr³⁺ is oxidized to Cr⁶⁺ during corrosion.
3. The accuracy of the thermodynamic calculation of FactSageTM software was verified for the MgO-NaCl system by analyzing the results of thermodynamic experiments. Combined with the author's previous research work, high-content MgO refractories can solve the corrosion problem of refractories caused by NaCl and KCl.

Author Contributions: Conceptualization, Y.Z., G.C. and C.D.; Software, Y.Z.; Validation, Y.Z. and L.L.; Formal Analysis, Y.Z., J.Z. and L.L.; Investigation, Y.Z. and L.L.; Resources, Y.Z. and C.D.; Data Curation, Y.Z., T.Z. and X.W.; Writing—Original Draft Preparation, Y.Z., F.L. and G.C.; Writing—Review and Editing, Y.Z. and G.C.; Visualization, Y.Z. and G.C.; Supervision, Y.Z. and C.D.; Project Administration, Y.Z. and G.C.; Funding Acquisition, Y.Z. All authors have read and agreed to the published version of the manuscript.

Funding: This work was financially supported by the National Natural Science Foundation of China (52006072), Fundamental Research Funds for the Central Universities (2019MS032, 2018ZD08).

Conflicts of Interest: The authors declare no conflict of interest.

References

1. Prigent, P.; Bouchetou, M.L.; Poirier, J. Andalusite: An amazing refractory raw material with excellent corrosion resistance to sodium vapours. *Ceram. Int.* **2011**, *37*, 2287–2296. [\[CrossRef\]](#)
2. Xu, T.T.; Xu, Y.B.; Li, Y.W.; Sang, S.B.; Wang, Q.H.; Zhu, T.B.; Nath, M.; Zhang, B. Corrosion mechanisms of magnesia-chrome refractories in copper slag and concurrent formation of hexavalent chromium. *J. Alloy. Compd.* **2019**, *786*, 306–313. [\[CrossRef\]](#)
3. Hirata, T.; Morimoto, T.; Ohta, S.; Uchida, N. Improvement of the corrosion resistance of alumina-chromia ceramic materials in molten slag. *J. Eur. Ceram. Soc.* **2003**, *23*, 2089–2096. [\[CrossRef\]](#)
4. Bouchetou, M.L.; Poirier, J.; Arbelaez Morales, L.; Chotard, T.; Joubert, O.; Weissenbacher, M. Synthesis of an innovative zirconia-mullite raw material sintered from andalusite and zircon precursors and an evaluation of its corrosion and thermal shock performance. *Ceram. Int.* **2019**, *45*, 12832–12844. [\[CrossRef\]](#)
5. Wang, X.H.; Zhao, P.D.; Chen, J.W.; Zhao, H.Z.; He, K. Corrosion resistance of Al–Cr-slag containing chromium–corundum refractories to slags with different basicity. *Ceram. Int.* **2018**, *44*, 12162–12168. [\[CrossRef\]](#)
6. Hirata, T.; Akiyama, K.; Yamamoto, H. Corrosion resistance of Cr₂O₃–Al₂O₃ ceramics by molten sodium sulphate–vanadium pentoxide. *J. Mater. Sci.* **2001**, *36*, 5927–5934. [\[CrossRef\]](#)
7. Mahnicka-Goremikina, L.; Svinka, R.; Svinka, V. Influence of ZrO₂ and WO₃ doping additives on the thermal properties of porous mullite ceramics. *Ceram. Int.* **2018**, *44*, 16873–16879. [\[CrossRef\]](#)
8. Kujur, M.K.; Roy, I.; Kumar, K.; Chintiaah, P.; Ghosh, S.; Ghosh, N.K. Influence of ZrO₂ and WO₃ doping additives on the thermal properties of porous mullite ceramics. *Mater. Today: Proc.* **2018**, *5*, 2359–2366.
9. Wiedemeier, H.; Singh, M. Thermal stability of refractory materials for high-temperature composite applications. *J. Mater. Sci.* **1991**, *26*, 2421–2430. [\[CrossRef\]](#)
10. He, L.P.; Chen, D.C.; Shang, S.P. Fabrication and wear properties of Al₂O₃–SiC ceramic coatings using aluminum phosphate as binder. *J. Mater. Sci.* **2004**, *39*, 4887–4892. [\[CrossRef\]](#)
11. Yoshikawa, M.; Iida, E.; Shikama, H.; Inoue, K. Application of chrome-free bricks for incinerated-ash melting furnaces. *J. Tech. Assoc Refract.* **2005**, *25*, 232. [\[CrossRef\]](#)
12. Moda, J.; Tanaka, K.; Kitamura, S. Chrome-free castables for waste melting furnaces. *J. Tech. Assoc Refract.* **2008**, *28*, 204–209.
13. Fan, R.D.; Zhao, H.Z.; Zhang, H.; Zhao, P.D.; Chen, J.W.; Wang, X.H. Effect of partial substitution of alumina-chromium slag for Al₂O₃ on microstructures and properties of Al₂O₃–SiC–C trough castables. *Ceram. Int.* **2019**, *45*, 11204–11215. [\[CrossRef\]](#)
14. Igabo, K.; Sakida, S.; Benino, Y. Development of Cr-free refractories for high temperature municipal waste incinerators. *J. Tech. Assoc Refract.* **2008**, *28*, 125.
15. Gehre, P.; Aneziris, C.G.; Veres, D.; Parr, C.; Fryda, H.; Neuroth, M. Improved spinel-containing refractory castables for slagging gasifiers. *J. Eur. Ceram. Soc.* **2013**, *33*, 1077–1086. [\[CrossRef\]](#)

16. Ren, B.; Li, Y.W.; Nath, M.; Wang, Q.H.; Xu, Y.B. Enhanced alkali vapor attack resistance of bauxite-SiC refractories for the working lining of cement rotary kilns via incorporation of andalusite. *Ceram. Int.* **2018**, *44*, 22113–22120. [[CrossRef](#)]
17. Bianco, R.; Jacobson, N. Corrosion of cordierite ceramics by sodium sulphate at 1000 °C. *J. Mater. Sci.* **1989**, *24*, 2903–2910. [[CrossRef](#)]
18. Stjernberg, J.; Olivas-Ogaz, M.A.; Antti, M.L.; Ion, J.C.; Lindblom, B. Laboratory scale study of the degradation of mullite /corundum refractories by reaction with alkali-doped deposit materials. *Ceram. Int.* **2013**, *39*, 791–800. [[CrossRef](#)]
19. Boris, R.; Antonovič, V.; Keriene, J.; Stonys, R.; Kudžma, A.; Zdanevičius, P. Study of alkali resistance of refractory materials used in boilers operating on wood fuel. *Refract. Ind Ceram.* **2017**, *57*, 651–654. [[CrossRef](#)]
20. Gregurek, D.; Schmidl, J.; Reinharter, K.; Reiter, V.; Spanring, A. Copper anode furnace: Chemical, mineralogical and thermo-chemical considerations of refractory wear mechanisms. *J. Miner. Metals Mater. Soc.* **2018**, *70*, 2428–2434. [[CrossRef](#)]
21. Stjernberg, J.; Lindblom, B.; Wikström, J.; Antti, M.L.; Odénc, M. Microstructural characterization of alkali metal mediated high temperature reactions in mullite based refractories. *Ceram. Int.* **2010**, *36*, 733–740. [[CrossRef](#)]
22. Huang, R.; Qian, X.; Lv, X.D.; Liu, P.S.; Zhang, J.Z. Slag–refractory interactions during ilmenite smelting: Thermodynamic simulation and experimental data. *Refract. Ind Ceram.* **2018**, *59*, 134–139.
23. Zhao, Y.; Cheng, G.S.; Xiang, Y.; Long, F.; Dong, C.Q. Thermodynamic study of the corrosion of refractories by sodium carbonate. *Materials* **2018**, *11*, 2197. [[CrossRef](#)] [[PubMed](#)]
24. Zhang, Z.J. *Physical Chemistry of Materials*; Chemical Industry Press: Beijing, China, 2006.

Publisher’s Note: MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



© 2020 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<http://creativecommons.org/licenses/by/4.0/>).